

dihedral angle of 5·3 (3)°, such that the two halves of the chelate system are each more planar than the ligand as a whole. The carboxylate and peptide groups have retained their expected planarity, and the steric requirements associated with this retention of planarity in these groups and the formation of the Cu–N(8) bond probably precludes a more coplanar arrangement of the complete ligand framework. The copper atom, as in the trihydrate, is significantly displaced from the equatorial plane about 0·13 Å toward the loosely bound water molecule. In this respect, it is interesting that the copper lies nearly in the plane of the carboxylate half of the ligand (deviation 0·006 Å), while it lies substantially out of the plane of the peptide half of the ligand (deviation 0·091 Å).

The crystal packing in  $[(\text{glygly})\text{Cu}^{II}] \cdot 2\text{H}_2\text{O}$  is dominated by a series of intermolecular hydrogen bonds involving all of the acidic protons in the complex (Table 4). As noted in the structure of glycylglycinato-copper(II) trihydrate (Strandberg, Lindqvist & Rosenstein, 1961), the strongly coordinated water molecule  $[\text{O}(1)\text{H}_2]$  forms two strong hydrogen bonds while the loosely bound water molecule  $[\text{O}(2)\text{H}_2]$  is involved in a weaker set of hydrogen bonds.

There also seems to be some contribution to packing stability owing to the presence of weak dimerization of the complexes about centers of symmetry  $(\frac{1}{2}-x, \frac{1}{2}-y, -z)$  involving Cu and O(1) [Cu–O(1) distance 3·083 (1) Å]. This weak dimerization extends, in a very qualitative sense, the coordination of the copper from five to six.

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## Refinement of the Crystal Structure of $\text{Fe}_3\text{BO}_6$

BY ROLAND DIEHL AND GERNOT BRANDT

*Institut für Angewandte Festkörperphysik der Fraunhofer-Gesellschaft, D-7800 Freiburg, Breisgau, Eckerstrasse 4, Germany (BRD)*

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$\text{Fe}_3\text{BO}_6$  is orthorhombic with  $a = 10\cdot048$  (2),  $b = 8\cdot531$  (2),  $c = 4\cdot466$  (1) Å, space group *Pnma* (No. 62),  $Z = 4$ . It is isostructural with the mineral norbergite,  $\text{Mg}_3\text{SiO}_4(\text{OH})_2$  [White, Miller & Nielsen, *Acta Cryst.* (1965), **19**, 1060–1061]. X-ray intensities were collected on a four-circle automatic single-crystal diffractometer with Zr-filtered Mo  $K\alpha$  radiation. The structure has been refined from 1241 reflexions to  $R = 0\cdot031$ . Both structurally independent iron atoms are surrounded by six oxygen atoms forming distorted octahedra. Boron is exclusively tetrahedrally coordinated by oxygen.

### Introduction

The crystal structure of  $\text{Fe}_3\text{BO}_6$  (White, Miller & Nielsen, 1965) has been found to be isotopic with that of the mineral norbergite,  $\text{Mg}_3\text{SiO}_4(\text{OH})_2$  (Taylor & West, 1929). In the paper of White *et al.* (1965) only mean bond distances Fe–O (2·04 Å) and B–O (1·47 Å)

are reported because of the limited accuracy of the oxygen coordinates which resulted from an overlap of the oxygen positions in an  $h0l$  electron density projection. Isotropic temperature factors are given: 0·52 Å<sup>2</sup> for the iron atoms, 0·90 Å<sup>2</sup> for the lighter atoms.  $\text{Fe}_3\text{BO}_6$  is a so-called easy-axis ferrimagnet with a Néel temperature of 235°C. In order to study the op-

tical and magnetic properties of this substance accurate structural information is required. We therefore decided to perform a full-matrix least-squares refinement of the  $\text{Fe}_3\text{BO}_6$  crystal structure including anisotropic thermal coefficients for all atoms.

Polycrystalline  $\text{Fe}_3\text{BO}_6$  was synthesized by reacting a mixture of  $\text{Fe}_2\text{O}_3$  and  $\text{B}_2\text{O}_3$  in a molar ratio of 1:1.5 at 850°C and subsequently leaching off the excess

$\text{B}_2\text{O}_3$ . Prismatic single crystals were obtained by chemical transport with chlorine in evacuated quartz ampoules (Diehl, 1975).

### Crystal data

Lattice constants of  $\text{Fe}_3\text{BO}_6$  were obtained by least-squares refinement from Guinier powder photographs

Table 1. Crystal data for  $\text{Fe}_3\text{BO}_6$

Present study	White <i>et al.</i> (1965)				
<i>a</i>	10.048 (2) Å	10.05 (1) Å	$\rho_{\text{exp}} = 4.78$ (2) g cm <sup>-3</sup>	$\rho_x = 4.76$ g cm <sup>-3</sup>	<i>Z</i> = 4
<i>b</i>	8.531 (2)	8.55 (1)			
<i>c</i>	4.466 (1)	4.470 (5)	F.W. 274.35	<i>F</i> (000) = 524	
<i>V</i>	382.82 Å <sup>3</sup>	384.1 Å <sup>3</sup>			

Table 2. Observed and calculated structure factors

H+0:0	10	175	-168	* H+1:1	5	17*	-7	9	302	20*	13	156	150	15*	-202	5	729	706	* H+2:1	11	191	-187	3	551	570	3	167	180								
4	121	121	* H+1:0	1	76	-72	6	54	46	10	367	-315	13	303	20*	14	56	17*	178	337	337	12	598	623	7	126	140									
4	1223	-766	* H+1:0	3	230	-265	8	215	-207	12	677	-777	H+0:2	8	201	192	1	14*	12	225	-255	13	427	434	5	244	265	5	220	220						
6	608	-766	* H+1:0	3	109	-149	9	13*	-13	13	143	-143	H+2:1	9	412	-403	2	145	-145	3	136	134	H+0:5	7	465	-476	7	105	116							
7	103	52	* H+0:2	505	46	-505	13	109	-149	13	143	-143	H+2:1	1	185	-181	1	860	806	11	220	219	209	265	370	352	-330	8	186	205						
10	259	259	* H+0:2	513	-508	5	973	522	110	130	143	-143	H+2:1	2	978	860	2	508	555	13	111	111	105	5	329	-320	6	422	-403							
12	605	-648	* H+0:2	645	-48	5	97	-16	12	55	-18	16	547	571	3	53	49	1	639	-101	12	55	57	5	329	-320	10	374	377							
14	16	16	* H+0:2	34	33	-33	16	16	-16	16	16	-16	H+2:1	3	348	-348	1	421	421	13	111	111	105	5	329	-320	11	117	122							
16	392	-105	* H+0:2	337	-333	9	223	-194	H+1:1	5	236	-236	6	112	-82	H+0:3	5	564	566	9	85	-79	5	232	228	H+1:6	7	147	-147	3	443	-440				
H+1:0	11	308	-290	* H+1:1	0	333	-322	1	453	20*	1	230	-219	2	131	-129	0	112	11*	10	347	-332	11	131	-131	7	245	235	H+0:7	1	135	125				
2	73	-30	* H+1:0	653	625	13	213	-204	1	630	635	2	216	-231	9	39*	-369	3	369	349	1	17*	11	89	100	H+1:6	10	146	-146	1	135	125				
4	731	647	* H+1:0	245	253	13	73	-75	3	792	791	3	508	-505	11	18*	-16	11	77	-76	3	92	-85	13	241	-230	10	156	156	H+1:6	153	57	-34	3	301	-316
6	603	558	* H+1:0	215	-211	8	87	90	4	64	-66	5	397	-397	12	18*	-17	8	29	-26	3	288	-286	13	156	-156	H+1:6	164	57	-34	4	57	-57			
8	610	-192	* H+1:0	34	33	-33	16	16	-16	16	16	-16	H+2:1	3	349	-349	15	143	-143	2	130	-130	13	132	-132	H+1:6	6	145	139	3	352	-343				
10	307	275	* H+1:0	128	135	H+1:1	7	552	-551	8	470	-469	H+0:2	1	236	-240	2	14*	-14	7	52	-516	3	167	-174	7	314	362	H+1:6	7	314	-314	3	314	-314	
12	517	-45	* H+1:0	34	33	-33	16	16	-16	16	16	-16	H+2:1	4	236	-236	1	453	453	13	101	-101	5	495	-495	H+1:6	9	48	48	3	463	-463				
14	501	-35	* H+1:0	34	33	-33	16	16	-16	16	16	-16	H+2:1	9	208	196	1	451	-451	13	101	-101	5	495	-495	H+1:6	10	209	-207	3	463	-463				
16	365	-360	* H+1:0	0	534	-495	10	55	-39	11	16*	-16	H+2:1	0	851	-851	H+0:3	0	101	-101	13	238	-233	5	195	-195	H+1:6	10	209	-207	3	463	-463			
18	234	-219	* H+1:0	103	107	12	232	232	13	54	-67	2	91	-90	H+2:1	1	507	-496	1	453	453	13	352	-352	5	565	-566	H+1:6	10	209	-207	3	463	-463		
20	72	-72	* H+1:0	671	617	H+1:1	1	207	-207	17	49	-19	6	206	-206	4	76	-76	1	187	-72	2	200	-196	H+1:6	8	130	120	H+1:6	8	130	120	H+1:6	8	130	120
22	72	-72	* H+1:0	671	617	H+1:1	1	207	-207	17	49	-19	6	206	-206	4	76	-76	1	187	-72	2	200	-196	H+1:6	8	130	120	H+1:6	8	130	120				
24	406	-523	* H+1:0	6	371	19	1	207	-207	17	49	-19	6	206	-206	4	76	-76	1	187	-72	2	200	-196	H+1:6	8	130	120	H+1:6	8	130	120				
26	100	-102	* H+1:0	0	18*	25	6	11*	9	39	-31	12	11*	-11	H+2:1	0	246	-242	2	202	-196	1	18*	-6	8	252	-252	2	202	-196	H+1:6	8	130	120		
28	600	-581	* H+1:0	2	61	51	2	231	-237	4	161	-162	H+2:1	0	801	716	0	19*	23	3	288	-287	2	300	-297	H+1:6	8	130	120	H+1:6	8	130	120			
30	143	-101	* H+1:0	6	420	422	11	223	-235	6	148	-146	1	123	-123	0	98	-98	5	70	-57	13	18*	30	4	326	-326	11	191	-191	H+1:6	8	130	120		
32	626	677	* H+1:0	215	214	7	340	-335	2	213	-213	10	203	-193	H+2:1	10	17*	-22	6	165	-165	19	202	-197	H+1:6	8	130	120	H+1:6	8	130	120				
34	64	60	* H+1:0	14	18*	16	8	15	12	10	19*	-10	5	155	-155	H+2:1	1	212	-212	8	72	-72	H+0:4	7	333	-340	8	319	-316	H+1:6	8	130	120			
36	140	-176	* H+1:0	17	18*	16	8	15	12	10	19*	-10	5	155	-155	H+2:1	1	212	-212	8	72	-72	H+0:4	7	333	-340	8	319	-316	H+1:6	8	130	120			
38	240	-306	* H+1:0	17	18*	16	8	15	12	10	19*	-10	5	155	-155	H+2:1	1	212	-212	8	72	-72	H+0:4	7	333	-340	8	319	-316	H+1:6	8	130	120			
40	1240	-1205	* H+1:0	17	18*	16	8	15	12	10	19*	-10	5	155	-155	H+2:1	1	212	-212	8	72	-72	H+0:4	7	333	-340	8	319	-316	H+1:6	8	130	120			
42	524	-529	* H+1:0	17	18*	16	8	15	12	10	19*	-10	5	155	-155	H+2:1	1	212	-212	8	72	-72	H+0:4	7	333	-340	8	319	-316	H+1:6	8	130	120			
44	130	-111	* H+1:0	59	478	1	207	-207	17	49	-19	6	206	-206	4	76	-76	1	187	-72	2	200	-196	H+1:6	8	130	120	H+1:6	8	130	120					
46	111	-111	* H+1:0	0	18*	17	10	201	-201	4	164	-164	H+2:1	1	176	-176	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
48	220	-204	* H+1:0	2	92	89	1	13*	-11	2	207	-207	H+2:1	2	164	-164	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
50	137	-364	* H+1:0	5	391	5	344	-344	2	164	-164	H+2:1	9	127	-128	2	196	-184	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120				
52	285	-284	* H+1:0	4	102	101	4	164	-164	H+2:1	2	164	-164	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120									
54	129	-116	* H+1:0	16	15*	15	16	25	-25	3	368	-369	H+2:1	4	103	-103	4	164	-164	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120			
56	123	-116	* H+1:0	16	15*	15	16	25	-25	3	368	-369	H+2:1	4	103	-103	4	164	-164	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120			
58	101	-93	* H+1:0	6	116	95	7	131	-131	1	151	-151	H+2:1	5	231	-232	1	16*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
60	717	-717	* H+1:0	7	131	131	6	127	-127	8	72	-72	H+2:1	6	156	-156	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
62	188	-2283	* H+1:0	10	91	121	216	216	-216	1	151	-151	H+2:1	7	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
64	893	-697	* H+1:0	6	165	619	619	3	437	-394	5	1233	-1194	H+2:1	8	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120					
66	555	-526	* H+1:0	13	61	-68	68	68	-68	6	213	-213	H+2:1	9	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
68	243	-243	* H+1:0	13	158	158	15	158	-158	1	151	-151	H+2:1	10	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
70	107	-194	* H+1:0	16	15*	15	16	25	-25	1	151	-151	H+2:1	11	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
72	776	-776	* H+1:0	17	16*	16	17	17	-17	1	151	-151	H+2:1	12	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
74	757	-757	* H+1:0	17	16*	16	17	17	-17	1	151	-151	H+2:1	13	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
76	705	-687	* H+1:0	16	15*	15	16	25	-25	1	151	-151	H+2:1	14	207	-207	1	18*	-17	1	204	-204	H+1:6	8	130	120	H+1:6	8	130	120						
78	966	-914	* H+1:0	13	116	-116	1	166	-166	0	1																									

taken with Ni-filtered Cu  $K\alpha_1$  radiation ( $\lambda=1.5405 \text{ \AA}$ ) using NaCl ( $a=5.6399 \text{ \AA}$ ) as an internal standard. They are listed in Table 1. The cell parameters reported by White *et al.* (1965) are given for comparison. Other relevant items are also given in Table 1.

Single crystals are black; the Mohs hardness is about 6. Systematic extinctions are  $0kl: k+l=2n+1$  and  $hk0: h=2n+1$ , indicating the space group to be either  $Pnma$  or  $Pn2_1a$ . From reflexion statistics using normalized structure factors (Karle & Karle, 1966) the structure was found to be centrosymmetric. Refinement was therefore carried out in the centrosymmetric space group  $Pnma$ . Fig. 1 gives an idealized drawing of the observed crystal morphology, [001] being the prism axis.

### Experimental

For X-ray intensity data collection a single crystal was ground to a sphere of radius  $R=208.5 \mu\text{m}$  in an air-operated crystal grinder. The sphere was mounted on a Nonius CAD-4 four-circle automatic single-crystal diffractometer with  $\kappa$  geometry. A total of 1241 independent reflexions up to  $(\sin \theta)/\lambda=0.91 \text{ \AA}^{-1}$  was measured with a scintillation counter (NaI:Tl) using Zr-filtered monochromatized Mo  $K\alpha$  radiation ( $\lambda=0.7107 \text{ \AA}$ ). Slow drift of the intensity of the primary beam was corrected by measuring the intensities of the standard reflexions 232, 042, and 123 every 20 measurements. The background-corrected integrated intensities were further corrected for Lorentz, polarization and absorption factors. The linear absorption coefficient  $\mu$  for Mo  $K\alpha$  radiation was found to be  $114.3 \text{ cm}^{-1}$  giving rise to a  $\mu R$  of 2.4. Absorption correction factors as a function of  $\theta$  were taken from *International Tables for X-ray Crystallography* (1959). Standard deviations were obtained only from counting statistics. 119 reflexions had an integrated intensity less than three times their standard deviation and were omitted as zeros from the structure refinement.

### Refinement of the crystal structure

The crystal structure of  $\text{Fe}_3\text{BO}_6$  was refined using the atomic coordinates given by White *et al.* (1965) as starting parameters.  $\text{Fe}^{3+}(1)$ ,  $\text{O}^{2-}(1)$ , and  $\text{O}^{2-}(2)$  are located at position 8(d),  $\text{Fe}^{3+}(2)$ ,  $\text{B}^{3+}$ ,  $\text{O}^{2-}(3)$ , and  $\text{O}^{2-}(4)$  at position 4(c) of the space group  $Pnma$  (No. 62). The atomic scattering factors for  $\text{Fe}^{3+}$  and  $\text{B}^{3+}$  were taken from Cromer & Waber (1965), those of  $\text{O}^{2-}$  from Baur (1965).  $\Delta f'$  and  $\Delta f''$  corrections for anomalous dispersion in the neutral atoms of Fe and O (Cromer & Liberman, 1970) were applied. A full-matrix least-squares refinement was performed using the program *CRYLSQ* of the X-ray 70 System (Stewart, Kundell & Baldwin, 1970). Weights were taken automatically as the reciprocals of the standard deviations of the observed structure factors. No additional weighting scheme was applied. In the case of

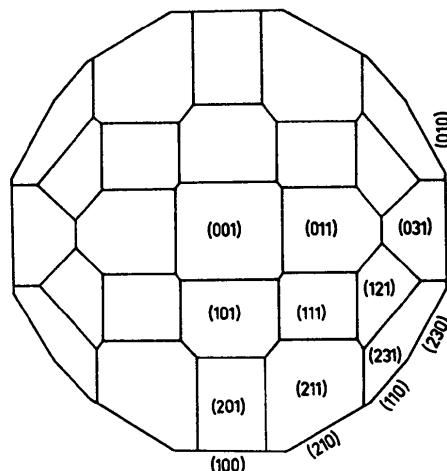


Fig. 1. Observed morphology of an  $\text{Fe}_3\text{BO}_6$  single crystal (idealized).

Table 3. *Atomic positional parameters*

		<i>x</i>	<i>y</i>	<i>z</i>
$\text{Fe}^{3+}(1)$	8(d)	0.12835 (4)	0.06466 (5)	0.98047 (9)
$\text{Fe}^{3+}(2)$	4(c)	0.41246 (6)	$\frac{1}{2}$	0.9749 (1)
$\text{B}^{3+}$	4(c)	0.21777 (4)	$\frac{1}{2}$	0.4106 (9)
$\text{O}^{2-}(1)$	8(d)	0.2856 (2)	0.1114 (2)	0.2647 (5)
$\text{O}^{2-}(2)$	8(d)	0.4654 (2)	0.0823 (2)	0.7351 (5)
$\text{O}^{2-}(3)$	4(c)	0.0813 (3)	$\frac{1}{2}$	0.2736 (7)
$\text{O}^{2-}(4)$	4(c)	0.2186 (3)	$\frac{1}{2}$	0.7333 (7)

Table 4. *Anisotropic temperature coefficients  $\times 10^2$*

	$U_{11}$	$U_{22}$	$U_{33}$	$U_{12}$	$U_{13}$	$U_{23}$
$\text{Fe}^{3+}(1)$	51 (2)	49 (2)	57 (2)	-1 (1)	-1 (1)	-5 (1)
$\text{Fe}^{3+}(2)$	51 (2)	54 (2)	52 (2)	0	6 (2)	0
$\text{B}^{3+}$	46 (13)	64 (14)	32 (13)	0	-5 (11)	0
$\text{O}^{2-}(1)$	56 (7)	45 (6)	54 (7)	10 (6)	-2 (6)	1 (6)
$\text{O}^{2-}(2)$	61 (7)	59 (7)	61 (7)	3 (6)	3 (6)	-22 (6)
$\text{O}^{2-}(3)$	43 (9)	58 (10)	53 (10)	0	-8 (8)	0
$\text{O}^{2-}(4)$	73 (10)	56 (10)	42 (10)	0	-4 (8)	0

Table 5. *Interatomic distances ( $\text{\AA}$ ) and angles ( $^\circ$ )*

Ionic charges are omitted for simplicity.

Coordinates of bonding atoms: for  $\text{Fe}(1)$ ,  $\text{Fe}(2)$ ,  $\text{B}$ ,  $\text{O}(1^1)$ ,  $\text{O}(2^1)$ ,  $\text{O}(3^1)$ , and  $\text{O}(4^1)$  see Table 3.

$\text{O}(1^{II})$	$x$	$y$	$1+z$
$\text{O}(1^{III})$	$\frac{1}{2}-x$	$\bar{y}$	$\frac{1}{2}+z$
$\text{O}(1^{IV})$	$x$	$\frac{1}{2}-y$	$1+z$
$\text{O}(1^1)$	$x$	$\frac{1}{2}-y$	$z$
$\text{O}(2^{II})$	$x$	$\frac{1}{2}-y$	$z$
$\text{O}(2^{III})$	$\frac{1}{2}-x$	$\bar{y}$	$\frac{1}{2}+z$
$\text{O}(2^{IV})$	$\bar{x}$	$y$	$\frac{1}{2}-z$
$\text{O}(3^{II})$	$x$	$y$	$1+z$
$\text{O}(3^{III})$	$\frac{1}{2}+x$	$y$	$\frac{1}{2}-z$
Fe(1) $\text{O}_6$ octahedron			
$\text{Fe}(1)-\text{O}(1^{II})$	2.069 (4)	Fe(2) $\text{O}_6$ octahedron	
$-\text{O}(1^{III})$	1.989 (4)	$\text{Fe}(2)-\text{O}(1^{II})$	$-\text{O}(1^{IV})$
$-\text{O}(2^{II})$	1.935 (4)	$-\text{O}(2^1)$	$-\text{O}(2^{II})$
$-\text{O}(2^{IV})$	1.904 (4)	$-\text{O}(3^{II})$	$-\text{O}(3^{III})$
$-\text{O}(3^{II})$	2.107 (3)	$-\text{O}(4^1)$	$-\text{O}(4^1)$
$-\text{O}(4^1)$	2.132 (4)		

Table 5 (cont.)

O(4 <sup>i</sup> )—Fe(1)—O(1 <sup>ii</sup> )	81.14 (17)	O(2 <sup>i</sup> )—Fe(2)—O(4 <sup>i</sup> )	88.33 (15)
O(4 <sup>i</sup> )—Fe(1)—O(3 <sup>ii</sup> )	81.84 (15)	O(4 <sup>i</sup> )—Fe(2)—O(2 <sup>ii</sup> )	95.52 (15)
O(4 <sup>i</sup> )—Fe(1)—O(1 <sup>iii</sup> )	97.40 (15)	O(2 <sup>i</sup> )—Fe(2)—O(1 <sup>ii</sup> )	95.52 (15)
O(4 <sup>i</sup> )—Fe(1)—O(2 <sup>iv</sup> )	92.67 (18)	O(1 <sup>iv</sup> )—Fe(2)—O(2 <sup>ii</sup> )	94.52 (15)
O(1 <sup>ii</sup> )—Fe(1)—O(3 <sup>ii</sup> )	68.66 (17)	O(2 <sup>i</sup> )—Fe(2)—O(3 <sup>iii</sup> )	94.52 (15)
O(1 <sup>ii</sup> )—Fe(1)—O(1 <sup>iii</sup> )	96.61 (14)	O(3 <sup>iii</sup> )—Fe(2)—O(2 <sup>ii</sup> )	
O(1 <sup>ii</sup> )—Fe(1)—O(2 <sup>iii</sup> )	97.92 (15)	O(4 <sup>i</sup> )—Fe(2)—O(1 <sup>ii</sup> )	76.93 (15)
O(3 <sup>ii</sup> )—Fe(1)—O(2 <sup>iii</sup> )	90.76 (16)	O(4 <sup>i</sup> )—Fe(2)—O(1 <sup>iv</sup> )	
O(3 <sup>ii</sup> )—Fe(1)—O(2 <sup>iv</sup> )	93.74 (18)	O(1 <sup>iv</sup> )—Fe(2)—O(3 <sup>iii</sup> )	99.36 (16)
O(1 <sup>iii</sup> )—Fe(1)—O(2 <sup>iii</sup> )	90.23 (15)	O(3 <sup>iii</sup> )—Fe(2)—O(1 <sup>iv</sup> )	
O(2 <sup>iii</sup> )—Fe(1)—O(2 <sup>iv</sup> )	85.96 (16)	O(1 <sup>iv</sup> )—Fe(2)—O(1 <sup>iv</sup> )	65.56 (13)
O(1 <sup>iii</sup> )—Fe(1)—O(2 <sup>iv</sup> )	101.04 (15)	O(2 <sup>i</sup> )—Fe(2)—O(2 <sup>ii</sup> )	100.29 (17)
O(4 <sup>i</sup> )—Fe(1)—O(2 <sup>iii</sup> )	172.37 (16)	O(2 <sup>i</sup> )—Fe(2)—O(1 <sup>iv</sup> )	158.04 (15)
O(1 <sup>ii</sup> )—Fe(1)—O(2 <sup>iv</sup> )	161.93 (15)	O(1 <sup>iv</sup> )—Fe(2)—O(2 <sup>ii</sup> )	
O(3 <sup>ii</sup> )—Fe(1)—O(1 <sup>iii</sup> )	165.22 (17)	O(4 <sup>i</sup> )—Fe(2)—O(3 <sup>iii</sup> )	175.55 (20)
BO <sub>4</sub> tetrahedron			
B—O(1 <sup>i</sup> )	1.504 (6)	O(1 <sup>i</sup> )—B—O(3 <sup>i</sup> )	103.58 (37)
—O(1 <sup>v</sup> )		O(3 <sup>i</sup> )—B—O(1 <sup>v</sup> )	
—O(3 <sup>i</sup> )	1.494 (9)	O(1 <sup>i</sup> )—B—O(4 <sup>i</sup> )	115.50 (36)
—O(4 <sup>i</sup> )	1.439 (9)	O(4 <sup>i</sup> )—B—O(1 <sup>v</sup> )	
		O(1 <sup>i</sup> )—B—O(1 <sup>v</sup> )	102.44 (46)
		O(3 <sup>i</sup> )—B—O(1 <sup>v</sup> )	114.58 (55)

individual isotropic thermal motion a residual  $R = \sum |F_o| - |F_c| / \sum |F_o|$  of 0.054 was obtained. Refinement with anisotropic temperature coefficients and isotropic extinction correction with Larson's (1967) method resulted in an extinction factor of  $2 \times 10^{-3}$ , an absolute scale of 2.884, an  $R$  of 0.031 with zeros omitted, and  $R=0.035$  with them included. After the final refinement cycle the shifts of all positional and temperature parameters were less than 1% of the corresponding standard deviations. The difference Fourier map calculated with all reflexions was featureless. The  $F_o, F_c$  listing is given in Table 2.

### Results and discussion

The final atomic coordinates are listed in Table 3, the anisotropic temperature coefficients in Table 4. The lattice constants given in the first row of Table 1 and the atomic coordinates of Table 3 were used to calculate interatomic distances and angles in the Fe<sub>3</sub>BO<sub>6</sub> structure by means of the program *BONDLA* of the X-ray 70 System (Stewart *et al.*, 1970). The results are compiled in Table 5.

The crystal structure of Fe<sub>3</sub>BO<sub>6</sub> consists of a slightly distorted hexagonal close-packed arrangement of oxygen atoms. Close-packed oxygen layers are stacked along [001]. A relationship to the olivine structure is obvious. The 'Fe<sub>2</sub>BO<sub>4</sub>' part of the chemical formula forms 'olivine' regions in the structure which are connected in the [010] direction by means of 'FeO<sub>2</sub>' units

representing 'brucite' regions. For a more detailed description of the norbergite structure see Taylor & West (1929), but note the different setting of the space group which is given as *Pbnm*!

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